Coupling of excitons and defect states in boron-nitride nanostructures

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The signature of defects in the optical spectra of hexagonal boron nitride (BN) is investigated using many-body perturbation theory. A single BN-sheet serves as a model for different layered BN nanostructures and crystals. In the sheet we embed prototypical defects such as a substitutional impurity, isolated boron and nitrogen vacancies, and the divacancy. Transitions between the deep defect levels and extended states produce characteristic excitation bands that should be responsible for the emission band around 4 eV, observed in luminescence experiments. In addition, defect bound excitons occur that are consistently treated in our \emph{ab initio} approach along with the “free” exciton. For defects in strong concentration, the coexistence of both bound and free excitons adds substructure to the main exciton peak and provides an explanation for the corresponding feature in cathodo- and photoluminescence spectra.

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I. INTRODUCTION

Boron-nitride crystals and nanostructures are excellent wide band-gap materials for the realization of ultraviolet (UV) lasers.1 Defects formed during the BN synthesis or introduced as impurities can bind excitons and act as centers for their recombination. This leads to (desired or undesired) luminescence bands and losses in the photon frequency range of interest. In fact, recent experiments2–9 have given evidence that defects strongly effect the luminescence of BN-based materials. In order to achieve optimal optical properties, the relevant defects have to be identified such that methods can be devised to tune their concentration in the material.

Boron nitride (BN) is known to exist in three different crystals structures, namely hexagonal BN (h-BN), cubic BN (c-BN), and wurtzite BN (w-BN). Among these structures, h-BN is the stable one at room temperature and ambient pressure. Like graphite, it consists of stacked BN layers with a honeycomb structure. BN nanostructures10,11 ranging from single and multiwall nanotubes to BN fullerens and nanocones can be envisaged as having been tailored from a single BN sheet. Unlike in graphite, the partially ionic character of the BN bond results in a wide band gap of about 6.5 eV for bulk h-BN.12 The combination of a large gap with a strong electron-hole attraction makes the optical properties of BN nanostructures largely independent of the layer arrangement.13,14 A single layer of h-BN is thus a generic system for studying the optical properties of crystalline and nanostructured BN including the exciton-defect interaction.

Detailed experiments on the cathodo- and photoluminescence of h-BN crystals and BN nanotubes have been performed in the recent years.2–9,14 The origin of some of the features in the BN luminescence spectra remains unsettled. For instance, the occurrence of a defect-related emission band at 4 eV, composed of regularly spaced peaks compatible with phonon replicas,7,15 can be associated with a deep level impurity. However, the defect that is responsible for this deep level has not yet been identified. Furthermore, the emission band at 5.77 eV shows different subpeaks on the low energy side. The origin of its substructure is unclear, since the main peak can be understood as being due to the recombination of a Frenkel-type exciton12 with a degenerate recombination line.13,16 Interpretations of the experimental findings are based on different mechanisms: quasi-donor-acceptor pairs,6 excitons coupled with structural defects,9,15 breaking of the exact hexagonal symmetry (e.g., by the presence of defects),16 or a dynamical Jahn-Teller effect17 (symmetry breaking due to geometry relaxation in the excited state).

Theoretical works on defects in BN structures have so far focused on the electronic and structural properties of point defects.18–24 The abundance (related to the formation energy) and stability of different defects has been addressed with detailed density-functional theory (DFT) calculations.20,25–29 The position of deep levels within the fundamental band gap has been calculated on the DFT level. However, DFT calculation suffer from the well-known underestimated of the band gap. Therefore, besides the neglect of the strong excitonic effects in BN, (de)excitation spectra based on DFT energy levels may not be reliable enough for comparison with experimental spectra.

In the present paper we address the optical properties of point defects in BN going beyond the approximation of DFT. We focus on a substitutional carbon impurity on the nitrogen site, the isolated vacancies and the BN divacancy that are common defects in BN. We will demonstrate that defects are responsible for the emission lines around 4 eV. Their presence can also explain the experimentally observed splitting.
of the main exciton peak. The paper is organized as follows. In Sec. II we summarize the computation methods employed in this work. In Sec. III we present the ground state results and the quasiparticle (QP) band structures. In Sec. IV we investigate the role of defects on the optical properties and finally in Sec. V we compare our results with experiments.

II. COMPUTATIONAL METHODS

Density functional theory (DFT) within the local spin density (LSDA) or gradient corrected approximations (GGA) for the exchange-correlation energy has been widely used to study defects in semiconductors and insulators. Although DFT is, in principle, exact, it is limited to the ground state and as a closed expression for the exchange-correlation energy functional is unknown, approximations like the LSDA or GGA have to be employed. Albeit theoretically well-founded, these approximate functionals bear inherent limitations like the well-known band gap problem which concomitantly affect the prediction of defect levels, in particular in wide band gap insulators. In the last years, many-body perturbation theory within the GW approximation has become a valid gap insulators. In the last years, many-body perturbation theory within the GW approximation has become a valid alternative to predict defects properties in semiconductors and insulators starting from the DFT ground state. Moreover, in going beyond the ground state, the optical properties of defects and the host insulator can be addressed by including the electron-hole interaction in the excited state via the Bethe-Salpeter equation (BSE).

Here we use this approach to study the role of prototypical defects in the optical properties of BN nanostructures: a substitutional carbon impurity on the nitrogen sublattice (CN), the boron vacancy (VB), the nitrogen vacancy (VN), and the boron-nitrogen divacancy (VBN). Calculations proceed in three steps: first we obtain the DFT ground state employing the LSDA approximation and a plane wave pseudopotential method (see Ref. 38 for details). Second we evaluate the quasiparticle band structure within the G0W0 approach and third we address the optical spectra via the Bethe-Salpeter equation (BSE).

As we are studying an isolated monolayer employing three-dimensional periodic boundary conditions, the size of the quasiparticle energies and the energy of the electron-hole attraction will depend on the intersheet distance. In Ref. 13 we had shown that the value of the gap of the pure h-BN sheet is 0.6 eV higher for an intersheet distance of 80 a.u. (value which we are using in the calculations presented here). At the same time we had observed an increase of the exciton binding energy with increasing intersheet distance that almost exactly cancels the variation of the band gap. We expect a similar cancellation effect in the presence of defects but cannot exclude a minor effect of the intersheet distance on the absolute position of defect peaks.

III. DFT-GROUND STATE AND QUASIPARTICLE STATES

In this section we address the electronic and geometric structure of the four defects and then compare the DFT-electronic structure with the quasiparticle states obtained from G0W0 calculations.

In Fig. 1 the equilibrium geometry of the CN and the vacancies VN, VB, and VBN is shown along with the BN bond distances.
distances in the vicinity of the defect center. In the case of substitutional impurity all the first neighbor B atoms of CN undergo a small outward relaxation of 0.113 a.u. (4% of the BN-bond distance) with respect to the ideal lattice positions. The defect retains its threefold symmetry. In the case of VB and VN the outward displacement of the first-neighbor atoms is larger. More importantly, in agreement with Ref. 24, we find that a (pseudo) Jahn-Teller effect breaks the threefold symmetry slightly. In the divacancy VN case, the threefold symmetry is already reduced to a mirror symmetry by the defect configuration itself. We observed that the first N and B neighbors decrease their mutual distance upon relaxation, as shown in Fig. 1(d). The defect geometries found here are in good agreement with those reported in Refs. 18 and 20.

As established already in earlier work, CN and the vacancies possess localized defect levels in the band gap. Here we briefly review the electronic structure of the defects as obtained in DFT-LSDA and then compare it with the quasiparticle states obtained on the level of the G0W0 approximation where the DFT band gap error is corrected. Since the prediction of defect emission and absorption spectra hinges on the defect level positions, a correct prediction of these quantities is a crucial point for the investigation of optical properties.

The defect levels as obtained within the DFT-L(S)DA are reported in the left panel of Fig. 2. In the case of CN the defect levels arise from the three bonds of the substitutional carbon impurity with its boron neighbors. These bonds combine into two fully occupied states located within the valence band and a third half-occupied localized state within the band gap. As a result of the occupation with only one electron there is a spin-splitting of \( \approx 0.5 \text{ eV} \) between the occupied (spin-up) and unoccupied (spin-down) state.

The situation is more complex for the boron vacancy. In this case a twofold degenerate and a nondegenerate defect state are formed from the unpaired orbitals at the three nitrogen neighbors. In agreement with Refs. 20, 26, and 29, these defect levels are located within the band gap. In the neutral vacancy the twofold degenerate level is occupied by three electrons and the nondegenerate level remains empty. Due to the Jahn-Teller distortion a small splitting of the degenerate level arises in addition to the exchange-splitting of the half- and unoccupied states. VB behaves as a triple acceptor.

For VN we obtain a half-filled level in the band gap, and two unoccupied levels almost degenerate in energy with the conduction bands. Due to spin-splitting their counterparts of opposite spin are found as resonances in the conduction band. Finally, in the case of the divacancy VB, there are four orbitals originating from the two boron and nitrogen neighbors. Only three are located in the band gap, one being occupied by two electrons and the upper two remaining empty.

The above picture obtained from the Kohn-Sham eigenvalues gives a qualitative description of the defect levels in the band gap. The position of the states is, however, quantitatively affected by the LSDA-band gap error and the approximate treatment of exchange with this XC functional as discussed above. Furthermore, the ordering of occupied and unoccupied states on the Kohn-Sham level may be different from that of the “true” quasiparticle energy levels (see, e.g., Ref. 30). The G0W0 method yields quasiparticle energies and an improved description of exchange and screening effects. Due to the reduced dimensionality of the BN monolayer there is an incomplete screening of the Coulomb interaction, such that the quasiparticle corrections are expected to be quite large. We calculated the quasiparticle band structure for all the cases considered above. The calculated band gap is in agreement with the results of Ref. 13. In the right panel of Fig. 2 the position of the defect quasiparticle energies within the band gap is shown schematically. The G0W0 corrections to the LSDA levels depend on the orbital occupation and on the wave-function character.30 The exchange splitting between occupied and unoccupied defect levels is strongly enhanced and the ordering of defect levels within the band gap is altered in some cases. In all cases we find that occupied levels are pushed closer to the valence bands and unoccupied ones closer to conduction bands. For CN the spin-splitting is significantly increased from 0.5 eV to 3.1 eV. A large exchange splitting is also present in the case of VN. For VB all unoccupied levels are pushed up by more than 2 eV, depending on the character of the wave function, and the level order is modified as well.

![FIG. 2. (Color online) Quasiparticle energies of CN, VB, VN, and VN as obtained from LDA and G0W0 calculations: schematic representation of energy position with in the band gap. The filled arrows indicate levels occupied with electrons with spin up or down. Dotted arrows indicate unoccupied states (holes), correspondingly.](image-url)
In the case of V_{BN}, unoccupied levels are found close to the conduction band edge, while the occupied ones remain in their position close to the valence band edge. This underlines the fact that the divacancy contains both donor and acceptor levels. The general picture obtained from G_0W_0 shows the clear deficiencies of the DFT-LSDA in the description of defect level positions in the case of BN, in particular, for unoccupied states. Vested with the defect quasiparticle spectrum we now turn to the optical properties.

IV. OPTICAL PROPERTIES

In order to discern excitonic effects on the optical properties, we present first in Fig. 3 the optical spectra in the independent particle picture (IP)\(^{46}\) (i.e., without the inclusion of the electron-hole coupling). This yields an overview over relevant excitation mechanisms and later allows one to trace down the effect of the electron-hole coupling.

We start the discussion with the case of C_N. Comparing the corresponding panel of Fig. 3 with the schematic representation of the defect levels in the band gap (Fig. 2) one can identify the different absorption peaks. The first band at 3.4 eV is due to transitions form the extended states at the valence edge to the defect levels within the band gap; the other band at 4.4 eV involves extended states below the valence band edge and defect levels. Above 6 eV we find the bulk-like spectrum that is only slightly perturbed by the presence of the defect. For the boron vacancy V_B there are two bands at 3.5 eV and 4.7 eV that are associated with transitions from defect resonances in the valence band to the localized levels in the band gap and a series of small peaks around 5.9 eV originating from valence band to defect transitions. The defect-related excitation bands of V_N at 4.1 eV, in contrast to the previous cases, involves extended states near the conduction band edge as final states. Only the faint defect-related excitation band around 6.6 eV, not visible in the figure, originates from valence band states. Another faint peak around 5 eV is due to transitions among the defect levels in the band gap. Finally, for the V_{BN} divacancy, we find peaks between 5.6 eV and 6.0 eV due to defect-defect transitions and valence band to defect transitions, respectively.

We now turn to the spectra\(^{46}\) including the electron-hole interaction via the Bethe-Salpeter equation. These are shown in Fig. 4. The inclusion of the electron-hole interaction has essentially two effects on the defect-related spectra: (i) a general red-shift of the defect excitation energies due to electron-hole attraction and (ii) a coupling of the independent particle excitations. This coupling can give rise to bound electron-hole states, the excitons. In our case the BSE leads to a red-shift of defect-defect transitions in the range from 0.6 to 1.3 eV depending on the character and the localization of the wave function. A common feature of all spectra is the large peak around 5.5 eV which corresponds to the strongly localized exciton of pure h-BN\(^{12,13,42}\) which is composed of a large number of independent electron-hole pairs. In the presence of a defect, electron-hole coupling leads to defect bound excitons besides the “free” excitons.

In the case of C_N and V_N, as one can see from Fig. 4, an additional small peak appears just below the main excitons. In the case of V_{BN} the main exciton band has three subpeaks due to the strong defect field with reduced symmetry. In order to disentangle the nature of the different peaks, we analyze the amplitude of the electron-hole pairs that compose each of the exciton eigenstates \(\lambda\), as a function of the energy difference of the quasiparticle states of the corresponding electron and hole states:

\[
A^\lambda(\omega) = \sum_{\eta=\{eh\}} |\langle\eta|\lambda\rangle|^2 \delta(\omega - E_\eta).
\]

For brevity, we report \(A^\lambda(\omega)\) only for V_B and V_{BN} (see Fig. 5). The plots for the other defects show generally a similar behavior.

Comparing for V_B the electron-hole amplitudes of the four excitation energies with the schematic diagram of Fig. 2, we can distinguish three kinds of excitations: (i) transitions from valence bands or defect resonances to the defect levels, that generate the peaks at 4.2 eV, (ii) the main exciton at 5.6 eV that is entirely composed of extended states, and (iii) a peak at 5.4 eV generated by the coupling of bulk excitations with transitions from and to defect levels in the band gap. The C_N and V_N cases present a structure similar to the one discussed above. From these results it is possible to infer three important consequences: (i) transitions from defect levels to valence or conduction bands with an energy close to the band gap energy mix with the bulk excitations giving rise to an additional peak close to the main exciton; (ii) transitions between defect levels are renormalized by the electron-hole interaction but do not mix with the bulk excitations; (iii) the main exciton peak maintains a structure similar to the bulk system.

The excitonic band of the divacancy V_{BN} is distinct from the other spectra as the divacancy is a donor and acceptor states of low symmetry in the vicinity of the band edges. The direct transitions between acceptor and donor states give rise to the absorption peaks around 4.8 eV, while the resonant coupling of transitions between the extended conduction and valence band states with donor and acceptor states splits the
main exciton in three subpeaks respectively at 5.37, 5.43, and 5.5 eV (see Fig. 5).

The observed side peaks to the main excitonic peak are compatible with the fine structure observed in different luminescence experiments.5,6,15,17

V. CONCLUSION

We have investigated the role of defects on the optical properties of BN nanostructures. We have treated four different plausible defects in a single sheet of hexagonal BN: a substitutional carbon impurity, a boron vacancy, a nitrogen vacancy, and a BN divacancy. Quasiparticle effects strongly modify the defect-level positions within the band gap and increase the exchange splitting between occupied and unoccupied levels. Electron-hole interaction (treated on the level of the Bethe-Salpeter equation) not only leads to a strongly bound excitonic state in pure BN but also strongly renormalizes transitions from and to defect levels. This means that the peaks in the optical spectra which are related to “deep level impurities” are strongly affected by the \( e^- e^- \) and \( e^- h^+ \) interaction and can be reliably calculated only on the level of many-body perturbation theory (as opposed to the frequently used random-phase approximation using DFT wave functions and energies). Experimental luminescence spectra (as presented, e.g., in Ref. 15) display a prominent peak close to 4 eV with phonon replica on the low energy side. Comparing the four theoretical absorption spectra (Fig. 4) with the experimental luminescence spectrum, the best agreement is obtained for the Boron vacancy where the main absorption peak of the low energy regime is at 4.2 eV. The Stokes shift due to possible relaxation of the excited state is not included in our calculation and limits the comparison of experimental luminescence with

FIG. 4. (Color online) Optical absorption spectra calculated by the Bethe-Salpeter equation. In the left panels the absorption cross section is increased by a factor of ten in order to make the defect-related peaks visible. Spectra are shown with a broadening of 0.1 eV (to simulate a typical experimental broadening) and 0.005 eV (in order to investigate the fine structure of the spectrum).

FIG. 5. (Color online) Amplitude of the different electron-hole pairing entering in the BSE, for the boron vacancy and the BN divacancy, compared with the pure h-BN monolayer.
theoretical absorption spectra. In the case of a strong (0.7 eV) Stokes shift, also the BN divacancy (where a defect related absorption peak is at 4.8 eV) could be responsible for the experimental observation. The nitrogen vacancy (absorption peak at 4.5 eV) might also explain the experiment. However, in this case a second strong luminescence peak should also be observable below 3 eV which seems not to be the case in experiments. The nitrogen vacancy (absorption peak is at 4.8 eV) could be responsible for the Stokes shift, also the BN divacancy (where a defect related absorption peak is at 5.6 eV). This peak is doubly degenerate in the pure BN sheet. Coupling of the free exciton with one or several transitions from or to defect levels gives rise to a splitting of this peak and additional side peaks. The details of the splitting depend on the nature of the impurity but also on its periodic arrangement. The fine structure of the luminescence peak at 5.7 eV (with additional side peaks at 5.46 and 5.64 eV) can possibly be explained through our calculations which clearly display defect-related side peaks of the “free” excitonic peak. Furthermore, any kind of breaking of the exact hexagonal symmetry can lead to a splitting of the bright doubly degenerate “free” excitonic peak into two peaks and will also (for bulk h-BN) render the “dark” exciton (located 0.1 eV below the bright exciton) visible. For reasons of computational feasibility, we have performed calculations only on point defects. Other kinds of defects such as dislocations and grain boundaries lead similarly to a breaking of symmetry and/or to the formation of localized states and could thus as well explain the fine structure of luminescence spectra (one would expect a similar interaction of the “free” exciton with dangling-bond states). Indeed, Jaffrenou et al. observed that the luminescence peaks at 5.46 and 5.64 eV occur mostly at grain boundaries and dislocations where defect states due to dangling bonds are supposed to occur. The aim of our present study is thus not to unambiguously assign a certain type of defect to the observed luminescence spectra, but rather to show the general mechanism leading to the deep impurity levels and the formation of defect bound excitons. We also note that we did not study all possible point defects. A recent study of the chemical composition of an isolated h-BN sheet suggests the incorporation of oxygen atoms in the honeycomb lattice. The same study also demonstrated that substitutional carbon atoms occur mostly in pairs, occupying neighboring nitrogen and boron sites. Further studies are necessary to decide if the latter two types of defects can also occur during the synthesis of BN nanostructures or if they are due to the experimental conditions in the transmission electron microscope where vacancies in the h-BN sheet are induced by the electron bombardment and may be subsequently filled by a substitutional atom.

An alternative route to symmetry breaking was proposed by Watanabe et al. They suggested that the dynamical Jahn-Teller effect (splitting of the degenerate exciton by polaronic effects) could explain the excitonic splitting in hexagonal BN. Further studies are necessary to disentangle the two effects.

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All DFT calculations have been performed using a 10×10 supercell containing 200 atoms. The distance between the BN sheet was 0.0001 a.u. A single k-point was used. We used the exchange-correlation functional of Ref. 49. For the calculations, we used the PWSCF code with Troullier-Martins pseudopotentials and a plane-wave cutoff of 50 Ry. All the atomic structures considered in the paper correspond to the 1 sampling of the elementary cell (which corresponds to the 1×1 sampling of the 10×10 supercell) does not contain the high-symmetry point \( K \), the point with the minimum direct gap. Therefore, the reported LDA gap of 4.82 eV in Fig. 2 is slightly larger than the converged value 4.5 eV.\(^{11}\)

COUPLING OF EXCITONS AND DEFECT STATES IN . . .